



北京工业大学
BEIJING UNIVERSITY OF TECHNOLOGY



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Recent advance in Thermionic cathode

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Outline

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2 / La_2O_3 -Mo cathode

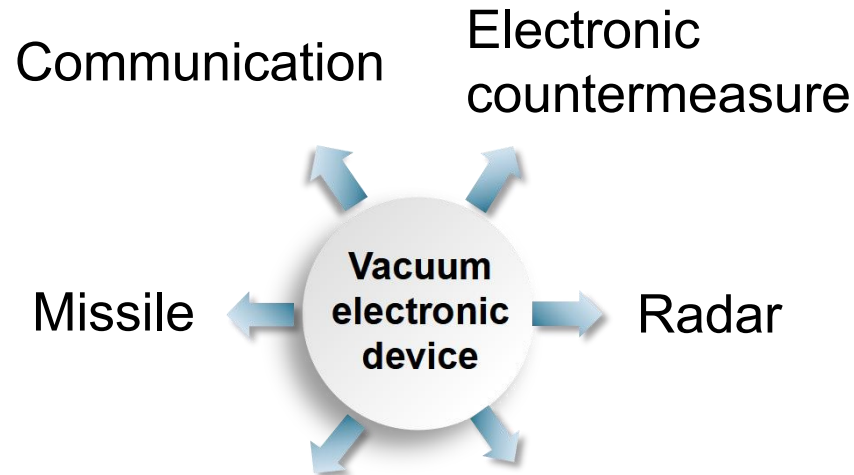
3 / Ba-W cathode

4 / Ir-W matrix scandate cathode

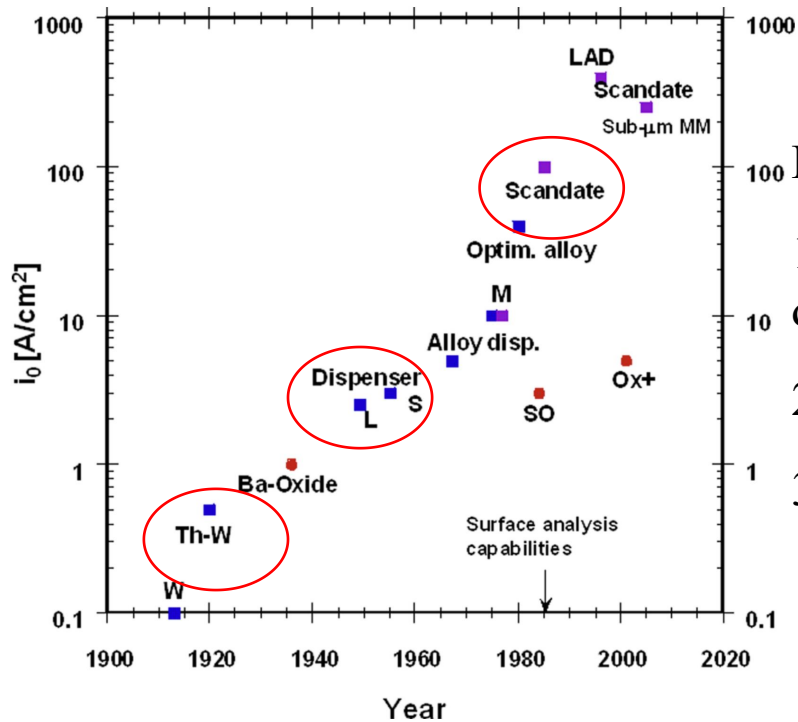
5 / Conclusion

Research Background

Vacuum electronic devices have a wide range of applications.



Research background



Thermionic cathodes

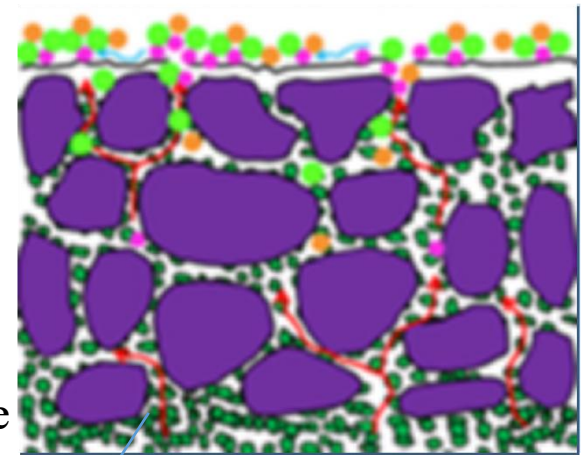
Dispenser cathode

1. Ba-W

dispensercathode

2. M type cathode

3. Scandate cathode



Porous tungsten

Aluminates

Filament cathode

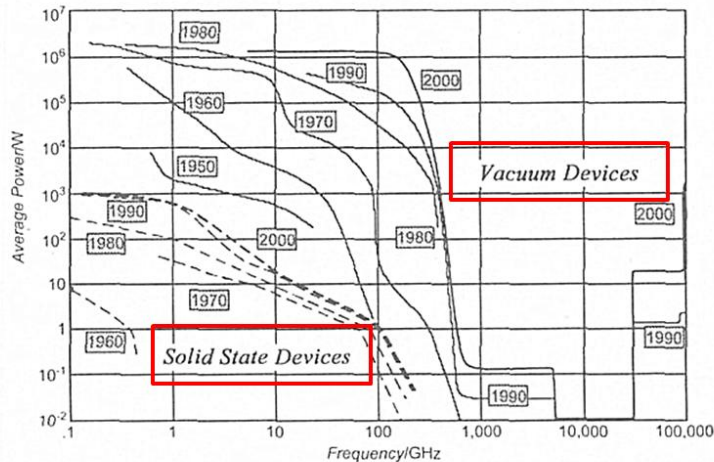


图1 微波真空电子器件和半导体器件单管功率输出

ThO₂-W cathode



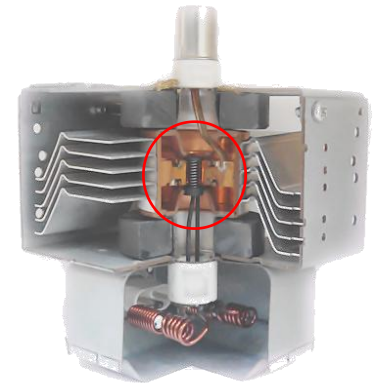
Created **in 1913** by Pintsch

High emission stability

微波炉



磁控管



1. Radioactivity
2. Unrecycled
3. Poor machinability
4. Brittle
5. High cost
6. High working temperature₅

Table Parameters of different refractory metals

Metal	Melting point(K)	Work function (eV)	A(A/cm ² K ²)	J=1A/cm ² Temperature (K) and evaporation rate (μ g/cm ² .s)
W	3650	4.52	75	2630 0.012
Mo	2890	4.24	51	2460 4
Ta	3300	4.19	55	2500 0.014
Nb	2770	4.01	30	2420 0.1
Re	3453	4.74	52	2780 0.5

RE→Th

Difficulties

RE₂O₃-Mo alloys

High Emission property

High concentration
RE₂O₃ (wt.%)
Bulk cathode: 30%
Filament cathode: 3-5%

Good ductility

Low concentration
RE₂O₃
(wt.%)
0-0.6%

~1000% deformation



Sintered
Φ16mm



Sintered
Φ16mm



Wires:
Φ0.5mm



Spring: twist
Φ0.5mm

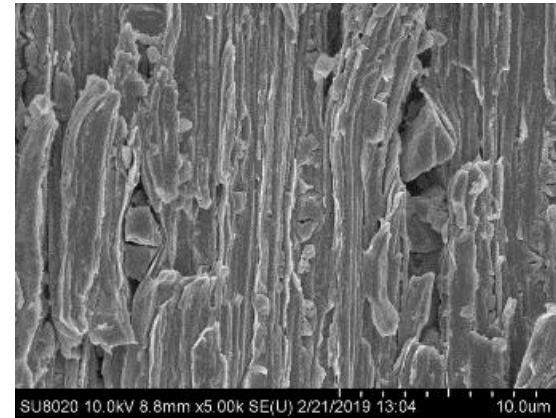
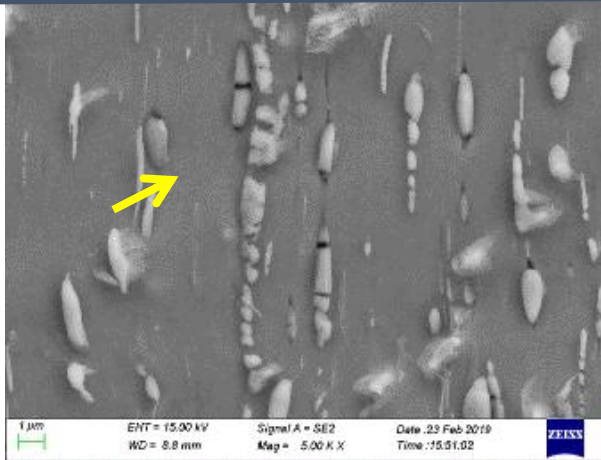


Weight percent/wt.%	STRAIN/% 25 °C	STRESS/MP a 25 °C
0	22.33	562.746
0.01(La ₂ O ₃)	23.97	626.196
0.03(La ₂ O ₃)	40.03	630.907
0.05(La ₂ O ₃)	32.72	645.592
0.08(La ₂ O ₃)	31.02	645.315
0.10(La ₂ O ₃)	24.47	645.315
0.50(La ₂ O ₃)	25.34	606.977
0.60(La ₂ O ₃)	37.58	850
1.00(La ₂ O ₃)	23.30	613.953
1.00(ZrO ₂)	17	700
1.50(La ₂ O ₃)	16.12	633.721
1.50(La ₂ O ₃)	35	
1.50(ZrO ₂)	18	760
2.00(La ₂ O ₃)	9.61	619.767
2.80(ZrO ₂ +La ₂ O ₃)	10	988

Competition between

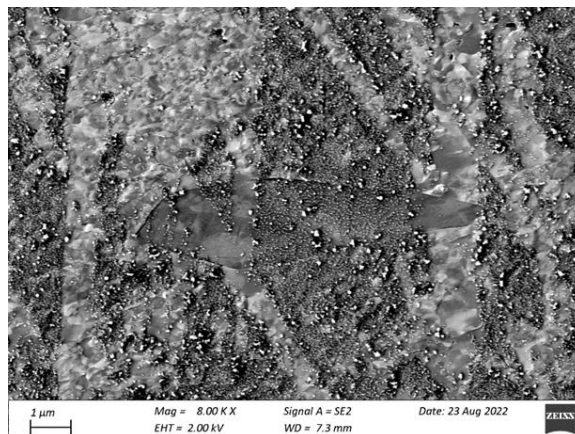
Emission property & ductility

Deformation of La_2O_3 -Mo alloys

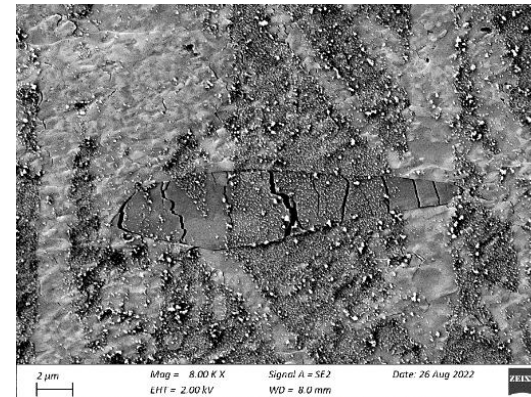


Poor ductility

Brittleness of ceramics: Ionic/Covalent bonding → High dislocation nucleation energy → cracks exist before deformation



strain=0%



strain=2%

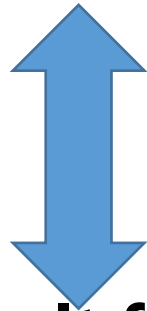
How to improve the ductility of alloys

Metal:

High ductility

Relatively easy for nucleation, multiplication and sliding of dislocations

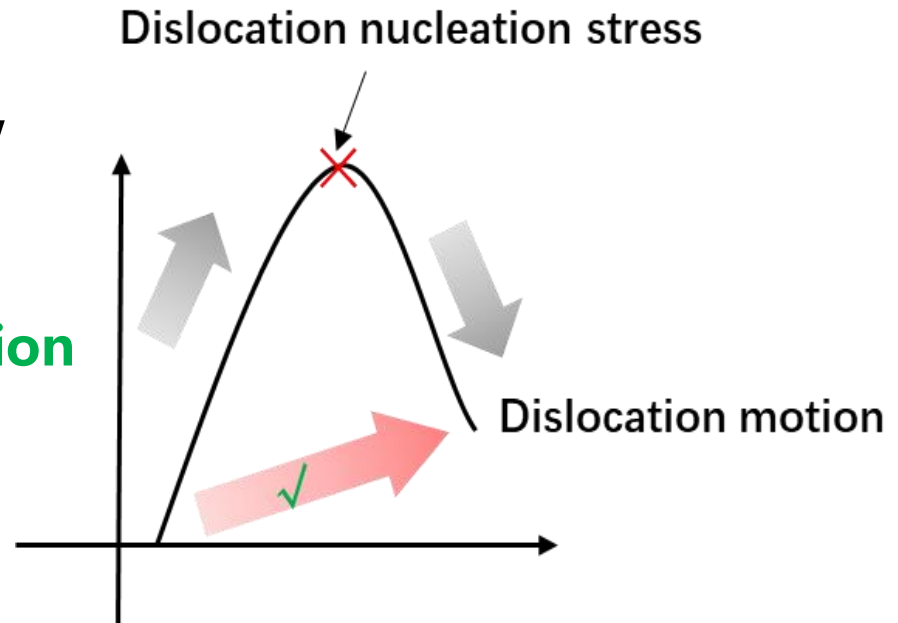
metal-ceramic
Interface



**Dislocation
corridor**

**Ceramics(Difficult for
dislocation nucleation)**

Borrow dislocations from metal
for dislocations propagation
and multiplication



L.R. Dong, JS Wang* et al, Science,
2024,385(6067),422-427

How to borrow dislocation for ceramics

To build the orderly-bonded interface

Condition 1 Condition 2

Chemical bonds

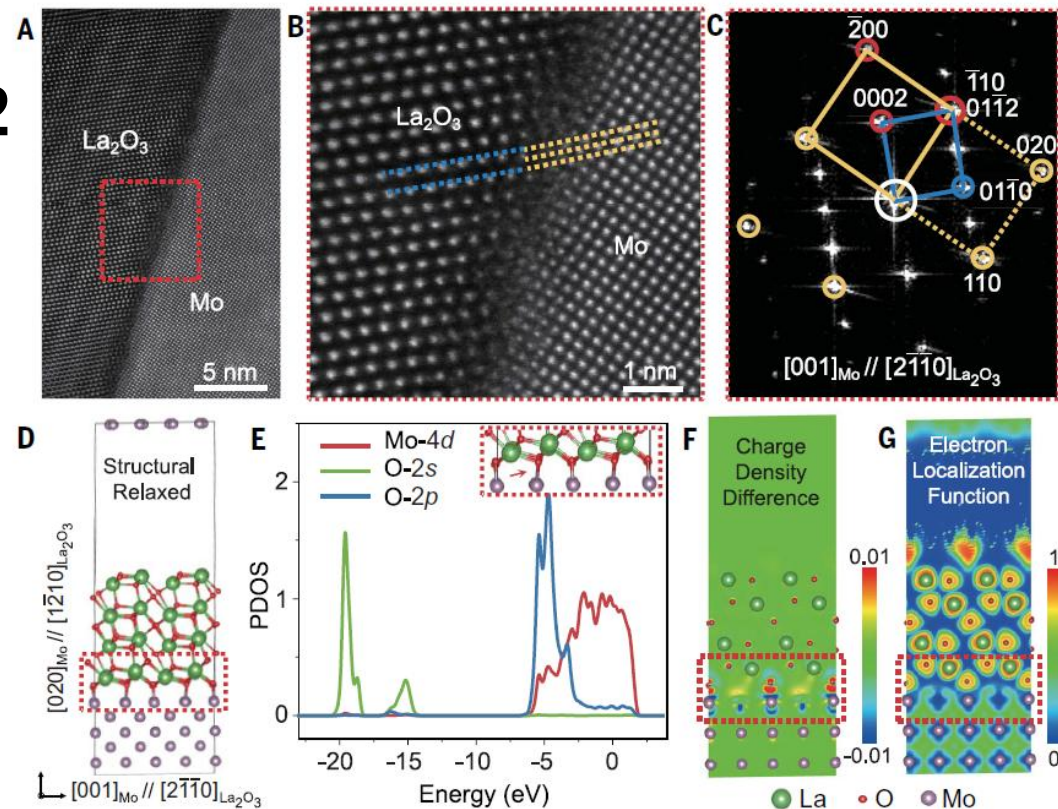
Ordered
interface



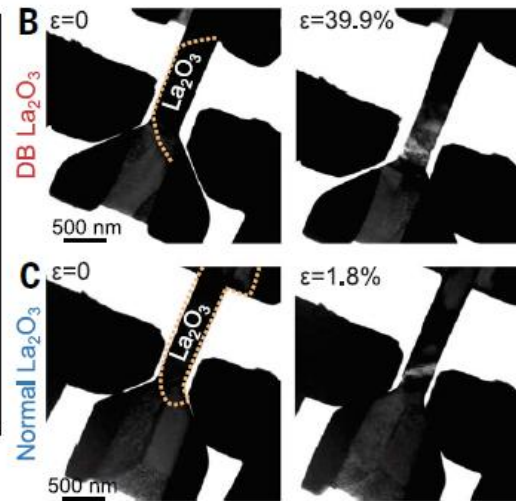
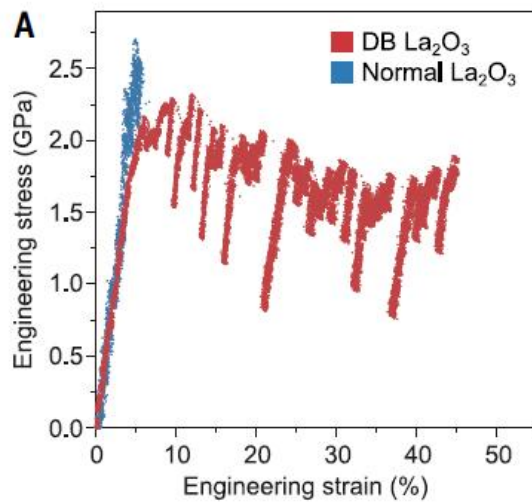
Strong



No dislocation
pile-up

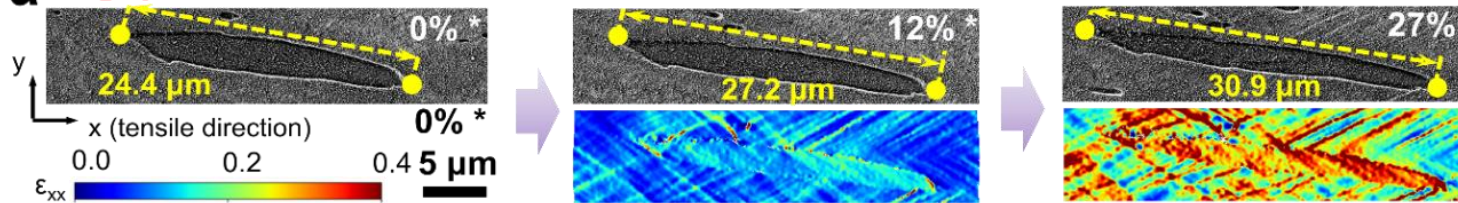


L.R. Dong, JS Wang* et al, Science, 2024,385(6067),422-427

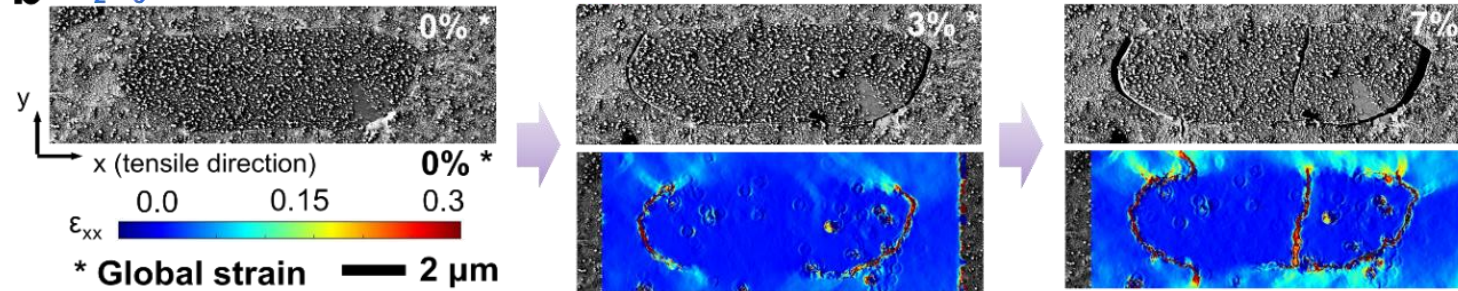


Engineering strain reach to 39.9%, Strength 2.3 Gpa

a La_2O_3 @Mo with tailored-interface

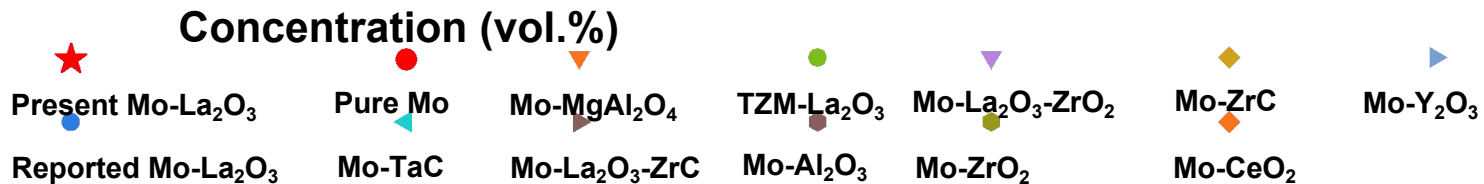
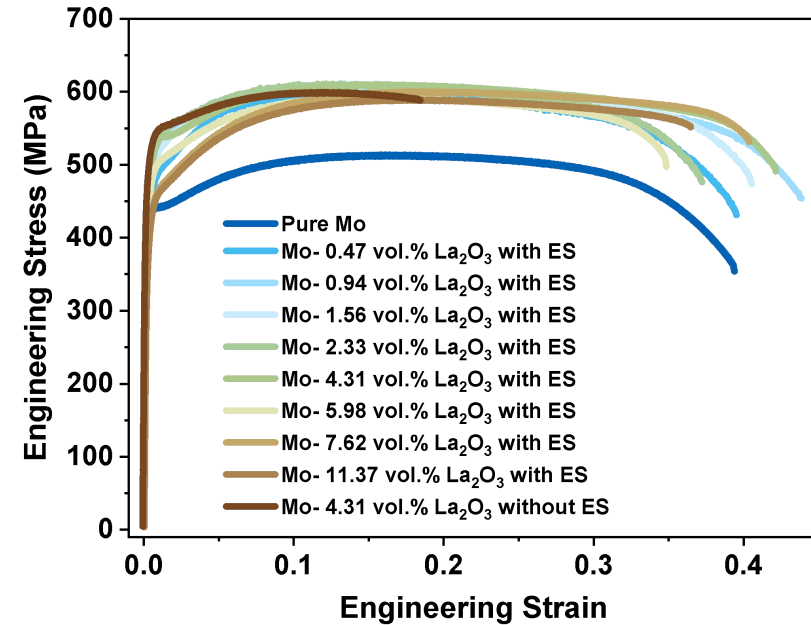
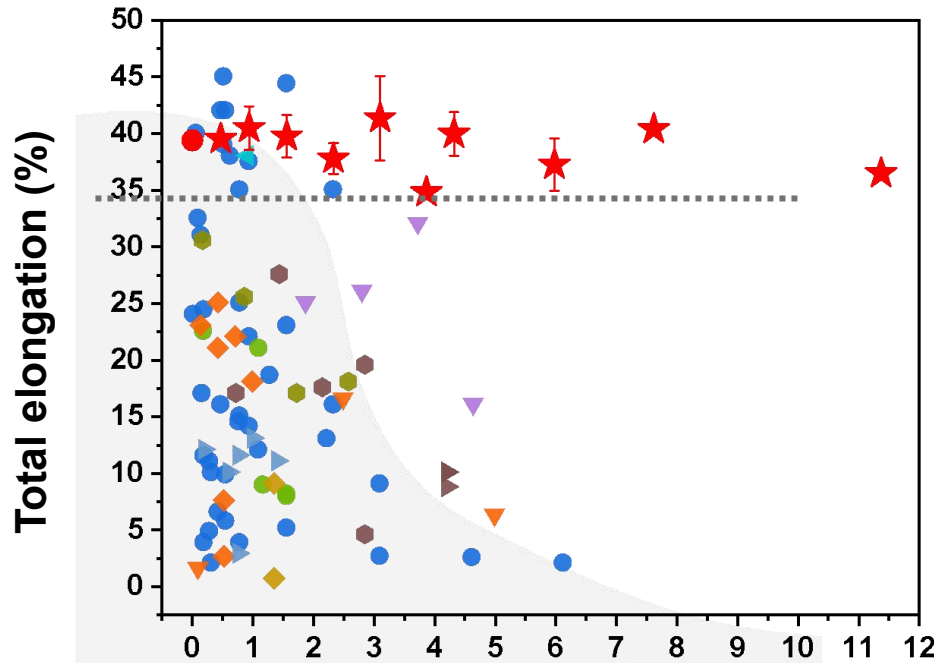


b La_2O_3 -Mo without tailored-interface



Tensile ductility of large ceramic particles can reach to 27.7%

The ductility of the Mo alloys with high concentration ceramic phase



The ductility of alloy keeps stable even with high concentration ceramic phase.

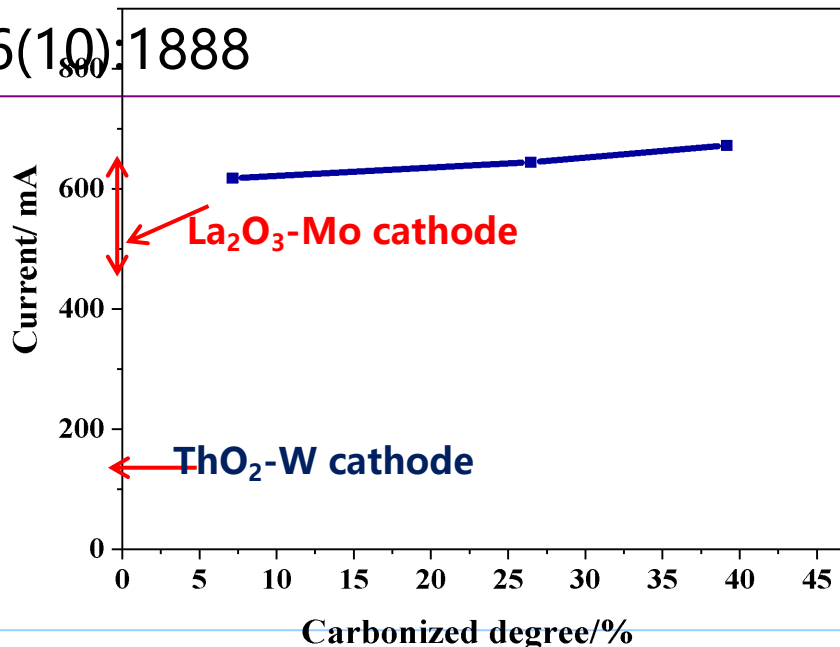
L.R. Dong, JS Wang* et al, Science, 2024,385(6067),422-427

Poor emission stability of La_2O_3 -Mo cathode

Ch.Buxbaum,Baden: Mo powder + 2% La_2O_3
powder $\rightarrow$ La_2O_3 -Mo
body $\rightarrow$ Plating(Pt) $\rightarrow$ Carburization(USP,4019081,1977)

Cathode is heated in the mixed gases of benzene and H_2

D M Goebel: Mo powder + 2% $\text{La}_2\text{O}_3 \rightarrow \text{La}_2\text{O}_3$ -Mo sintered body $\rightarrow$ Carburization, life 100h. Rev.Sci.Instrum,1985, 56(10):1888



La_2O_3 -Mo filament $\rightarrow$ carburization, life: 90-100h

La₂O₃-Mo cathode

Short lifetime

Poor emission stability?

High temperature evaporation of
 $3\text{Mo}_2\text{C} + \text{La}_2\text{O}_3 = 6\text{Mo} + 2\text{La} + 3\text{CO}$



Low
 temperature
 carburization?
 $2\text{Mo} + \text{C} = \text{Mo}_2\text{C}$



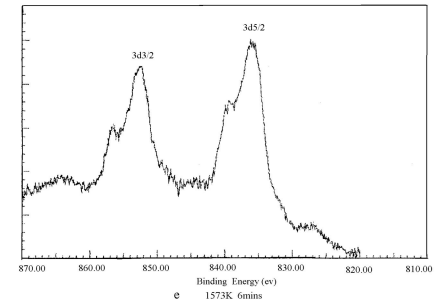
Kinetics



Carburization at lower
 temperature

Carburization
 temperature 1700°C

1350-1450°C



First stage:

$$\frac{\Delta W}{A} = 3.6668 \times 10^5 \exp\left(-\frac{2.4075 \times 10^5}{RT}\right)t$$

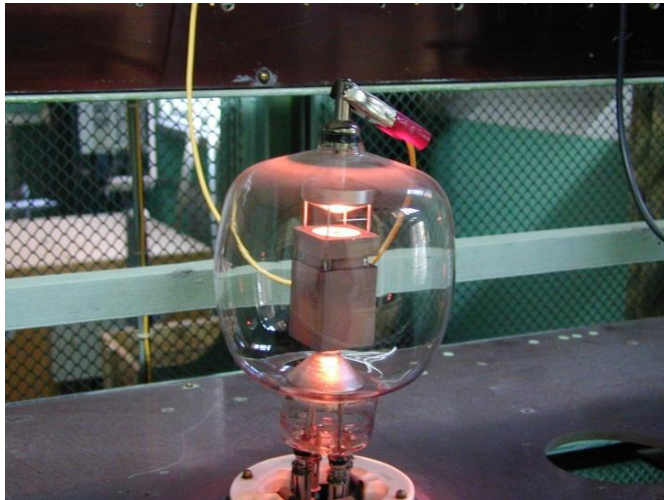
Second
 stage:

$$\left(\frac{\Delta W}{A}\right)^2 + 2.0517 \times 10^5 \exp\left(-\frac{1.9455 \times 10^5}{RT}\right)\left(\frac{\Delta W}{A}\right)$$

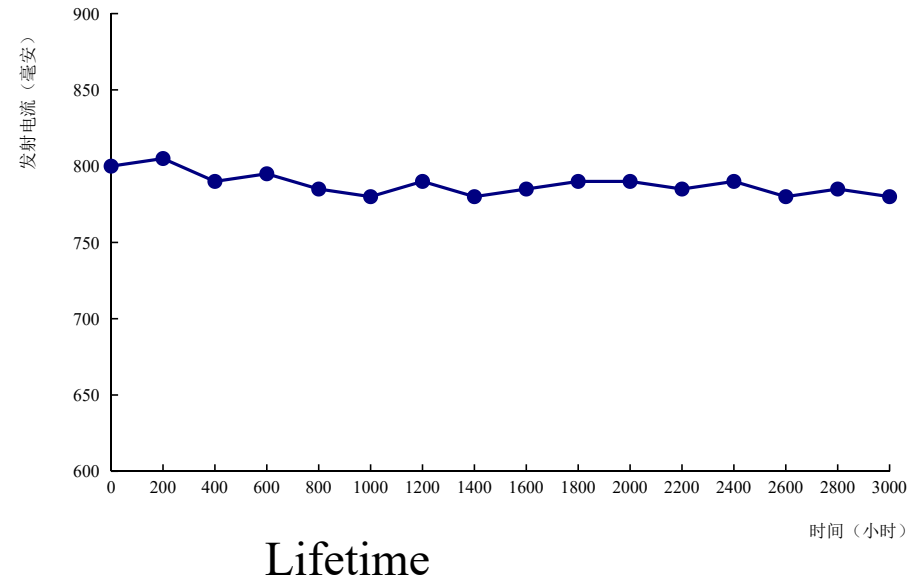
$$= 4.58678 \times 10^{10} \exp\left(-\frac{4.27273 \times 10^5}{RT}\right)t$$

Third stage:
 $\left(\frac{\Delta W}{A}\right)^2$

$$= 5.2782 \times 10^{10} \exp\left(-\frac{4.3114 \times 10^5}{RT}\right)t$$



FU6051 electron tube



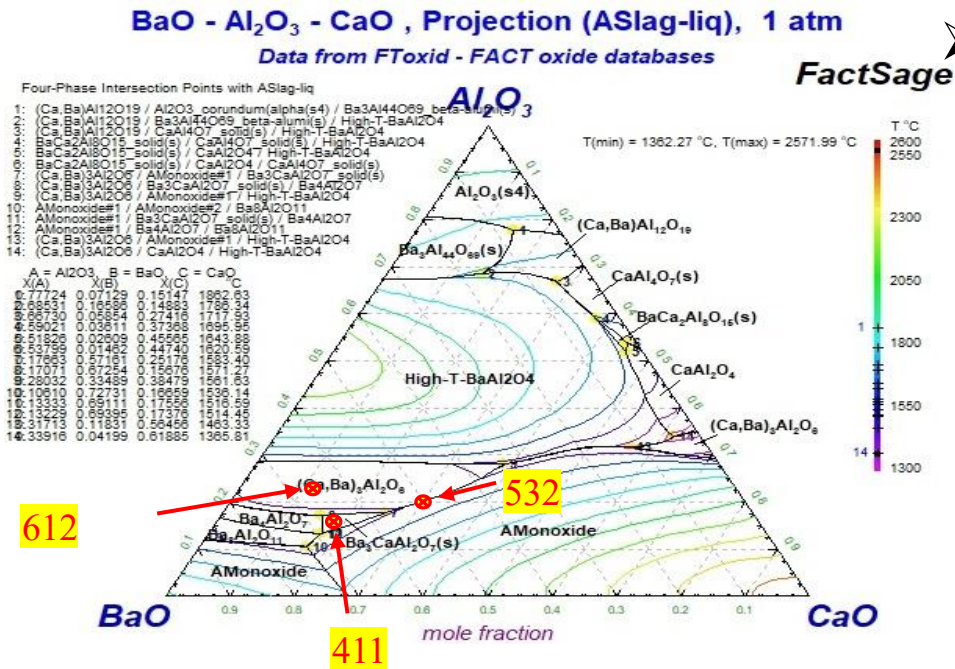
Lifetime: 100h→3000h
(technical standard 1000h)

Ba-W cathode

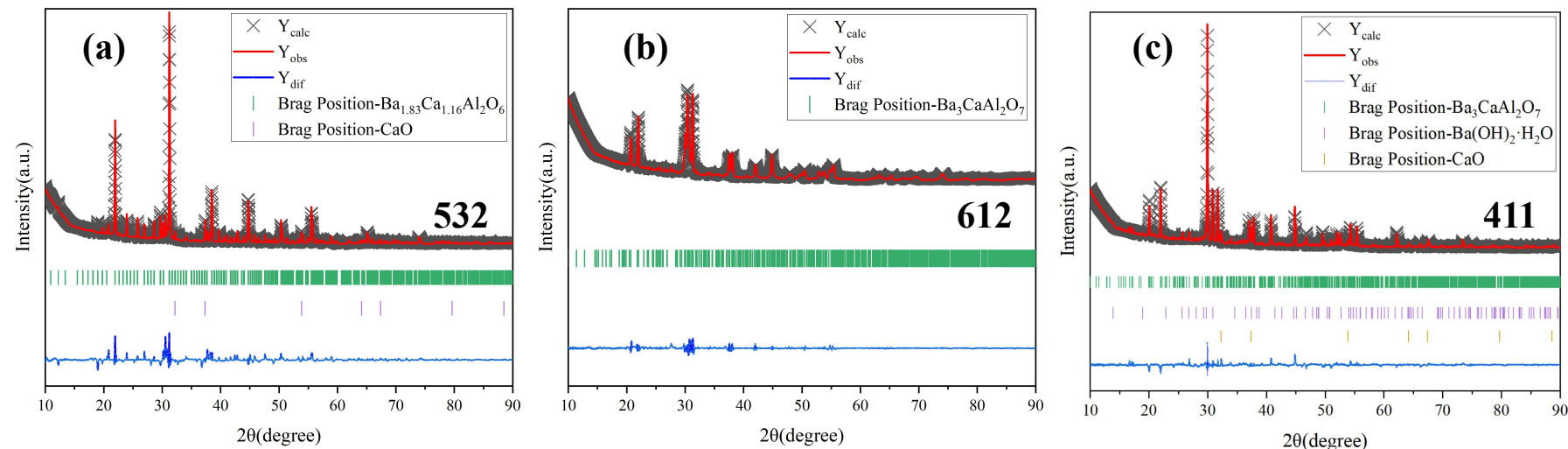
(Ba,Ca)aluminates

Type of aluminates: 532,612 and 411

Reaction process of the aluminates



Active phase and crystal structure



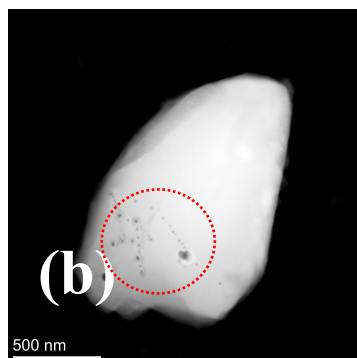
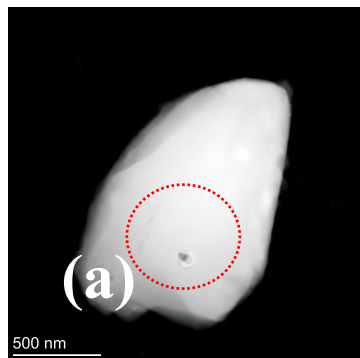
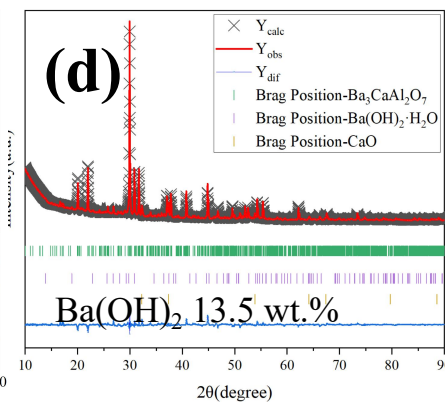
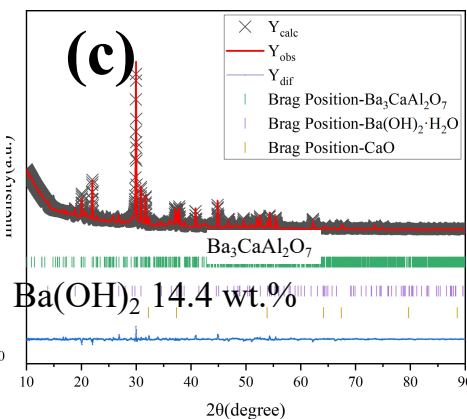
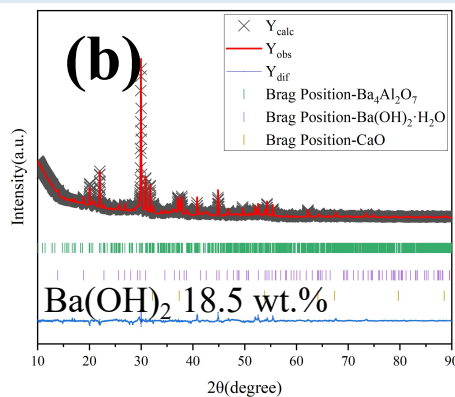
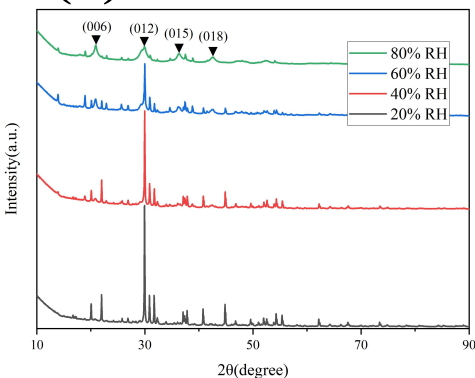
Sample	532	612	411
Data source	ICSD(246227)	PDF(48-0569)	ICSD(150769)
Phase	$\text{Ba}_{1.83}\text{Ca}_{1.17}\text{Al}_2\text{O}_6$	$\text{Ba}_3\text{CaAl}_2\text{O}_7$	$\text{Ba}_3\text{CaAl}_2\text{O}_7(\text{ss})$
Mass(wt.%)	88.8(5)	100	81.7(5)
Space group	$P2_13$	$Cmca$	$Cmca$
a(Å)	16.1857(9)	11.1360	8.0886(6)
b(Å)	16.1857(8)	27.6413	8.0866(4)
c(Å)	16.1857(7)	11.6731	26.5488(10)
V(Å ³)	4240.28(15)	3601.88952	1735.91(17)
$\alpha, \beta, \gamma(\text{deg.})$	90, 90, 90	90, 90, 90	90, 90, 90
Z	24	16	16
R _{wp} (%)	13.89	5.33	7.09

- 532: main phase $\text{Ba}_{1.83}\text{Ca}_{1.17}\text{Al}_2\text{O}_6(\text{ss})$, a small amount CaO
- 612: $\text{Ba}_3\text{CaAl}_2\text{O}_7$
- 411: $\text{Ba}_3\text{CaAl}_2\text{O}_7(\text{ss})$, a small amount of CaO and $\text{Ba}(\text{OH})_2$

Crystal structure of active phase

Formation of $\text{Ba}(\text{OH})_2$ in 411 under different humidity

(a)

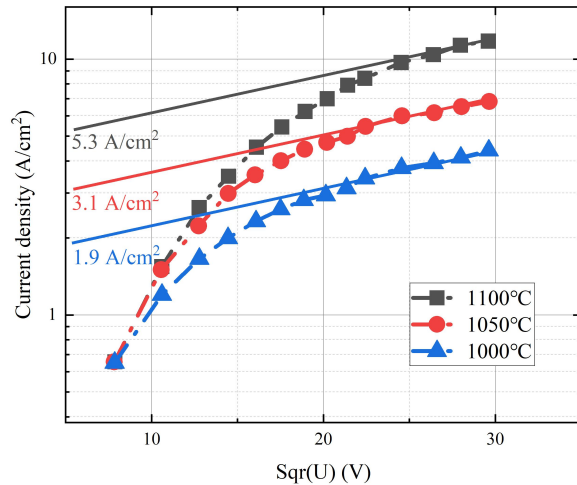


- 20-60% RH, Main phase $\text{Ba}_3\text{CaAl}_2\text{O}_7$ (ss)
- 80% RH, Main phase is hydrotalcite

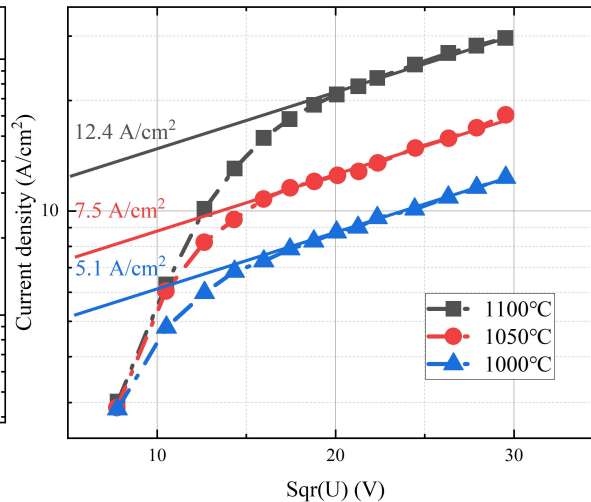
PXRD and Rietveld PXRD (b) 60% RH-411; (c) 40% RH-411; (d) 20% RH-411; TEM (e) before (f) after electron bombardment

Compared with 532 and 612, 411 is not stable and easy to absorb water to form $\text{Ba}(\text{OH})_2$

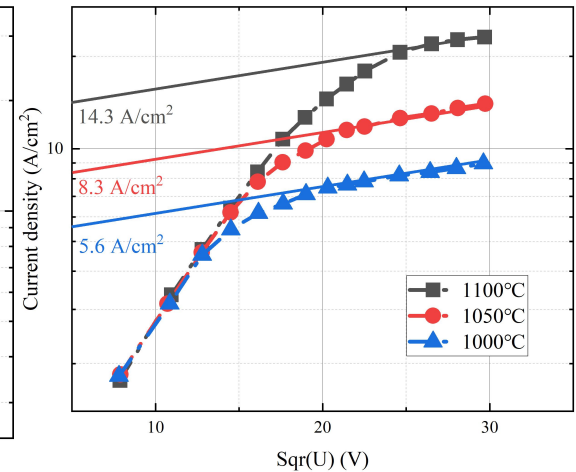
Emission property of the Ba-W cathode impregnated with different aluminates



(a) 532



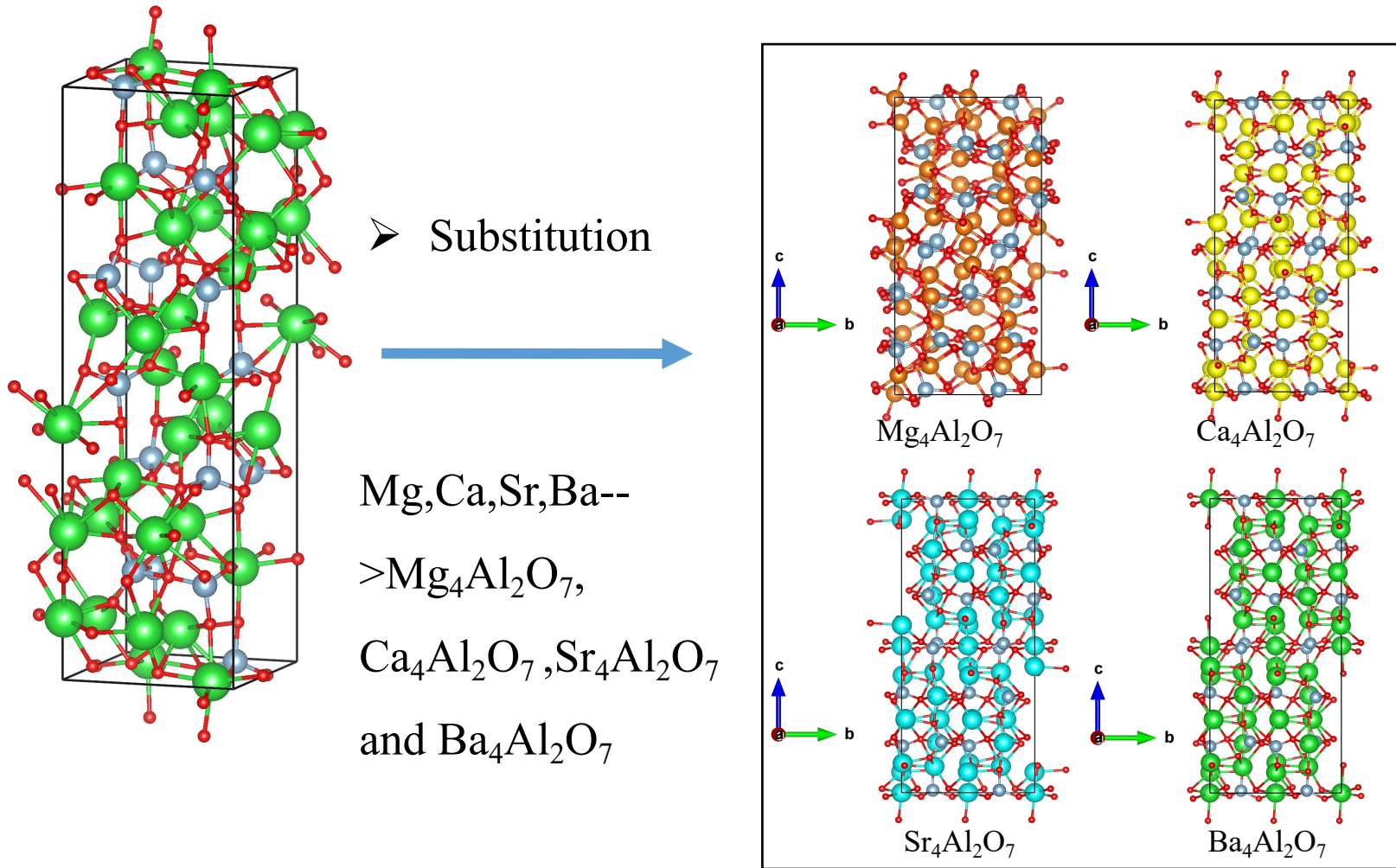
(b) 612



(c) 411

- Electron emission property: 411 > 612 > 532
- The contribution of different main phases: $\text{Ba}_3\text{CaAl}_2\text{O}_7(\text{ss}) > \text{Ba}_3\text{CaAl}_2\text{O}_7 > \text{Ba}_{1.83}\text{Ca}_{1.17}\text{Al}_2\text{O}_6(\text{ss})$

Crystal structure of $\text{Ba}_3\text{CaAl}_2\text{O}_7(\text{ss})$ -- $\text{M}_4\text{Al}_2\text{O}_7$



ICSD-No. 150769

$\text{M}_4\text{Al}_2\text{O}_7$

Fig.(a) Schematic daigram of $\text{Ba}_3\text{CaAl}_2\text{O}_7(\text{ss})$ initial $\text{M}_4\text{Al}_2\text{O}_7$

$M_4Al_2O_7$ phase

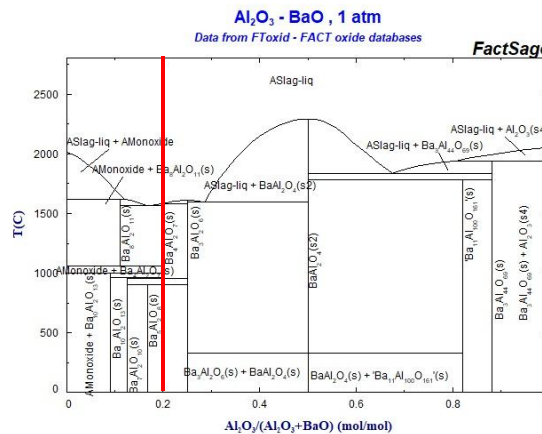
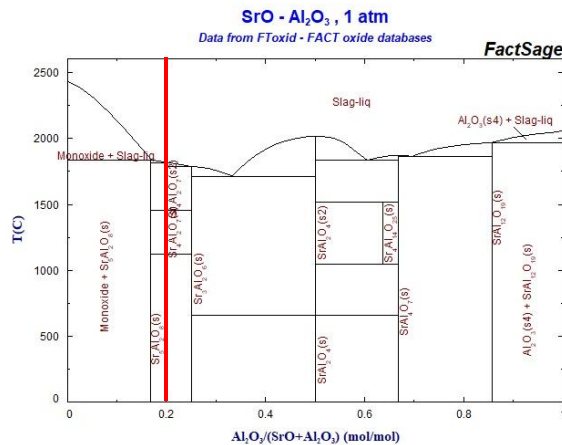
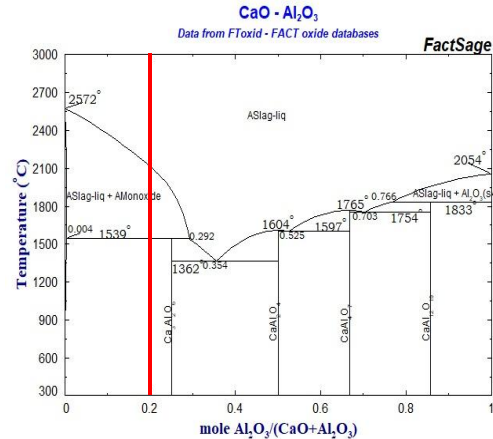
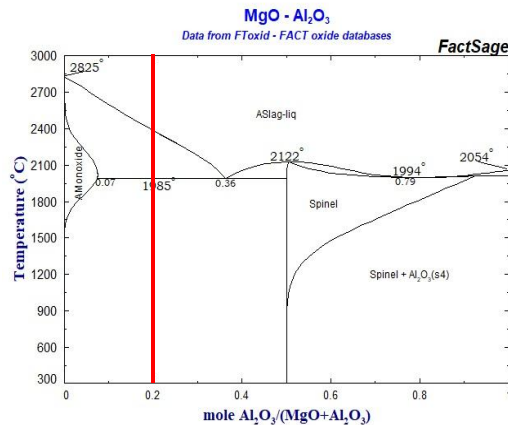
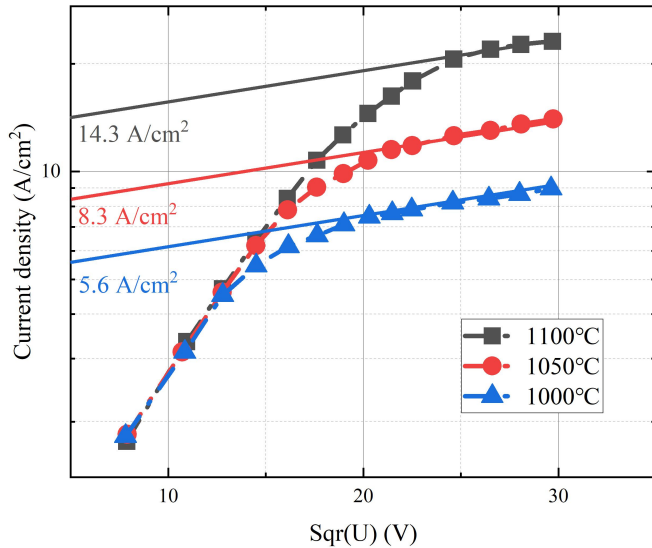


Fig. $MO \cdot Al_2O_3$ phase diagram

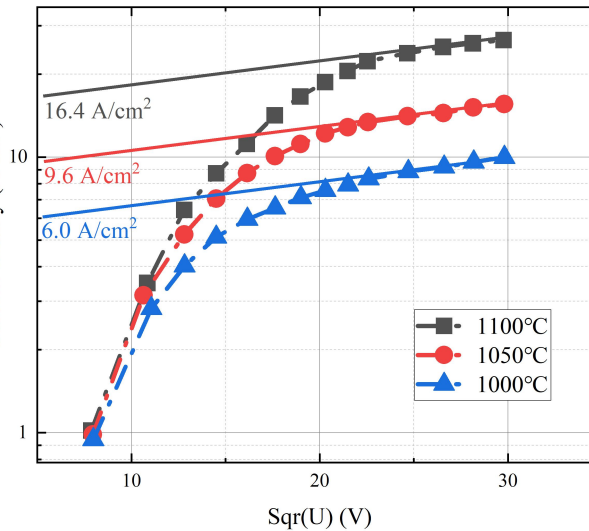
- $4MO \cdot Al_2O_3$ theoretical phase
- $MgO-Al_2O_3$: $MgO, MgAl_2O_4$
- $CaO-Al_2O_3$: $CaO, Ca_3Al_2O_6$
- $SrO-Al_2O_3$: $Sr_4Al_2O_7$
- $BaO-Al_2O_3$: $Ba_4Al_2O_7$
- $4MO \cdot Al_2O_3$ experimental results
- $MgO-Al_2O_3$: $MgO, MgAl_2O_4$
- $CaO-Al_2O_3$: $CaAl_2O_4, Ca_3Al_2O_6$
- $SrO-Al_2O_3$: $Sr_4Al_2O_7$
- $BaO-Al_2O_3$: $Ba_4Al_2O_7$

Sr and Ba exist as $M_4Al_2O_7$

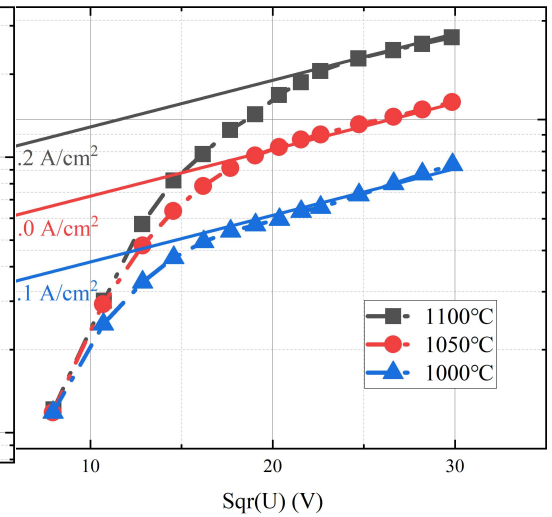
Electron emission property of the cathodes impregnated with Sr addition amount



(a) 0Sr



(b) 0.6Sr



(c) 1Sr

- Emission property: 0.6Sr-W > 0Sr-W > 1Sr-W
- Contribution of the main phase in cathode electron emission:
 $(\text{Ba, Sr, Ca})_4\text{Al}_2\text{O}_7 > \text{Ba}_3\text{CaAl}_2\text{O}_7(\text{ss}) > (\text{Ba, Sr})_4\text{Al}_2\text{O}_7$

Appropriate amount of Sr addition in the aluminates could increase the emission property of the cathode

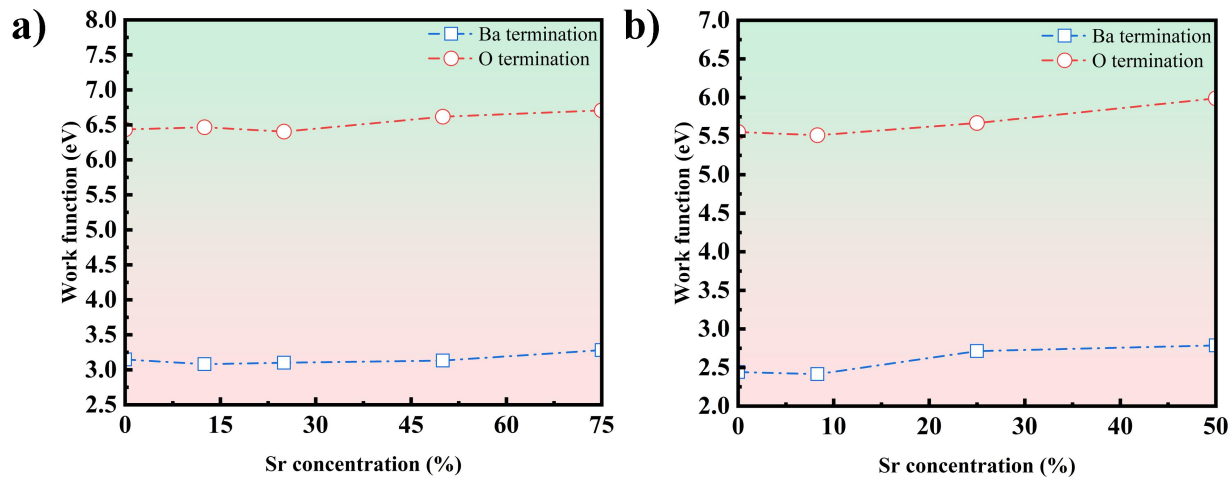
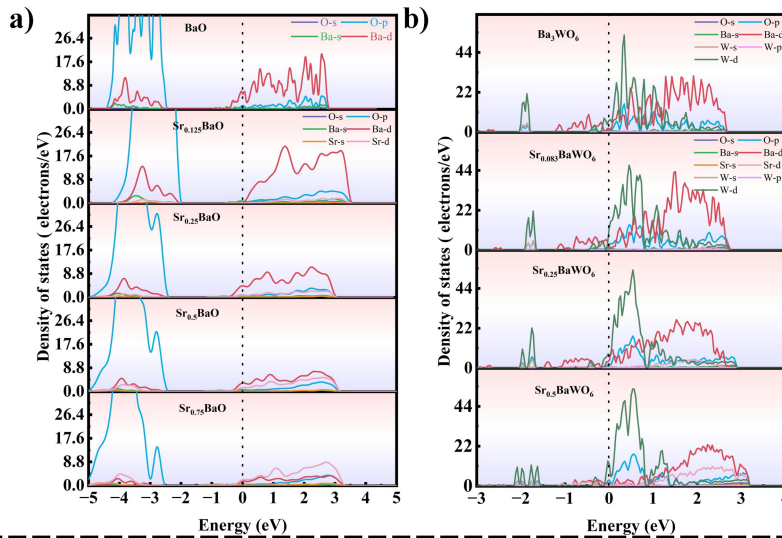
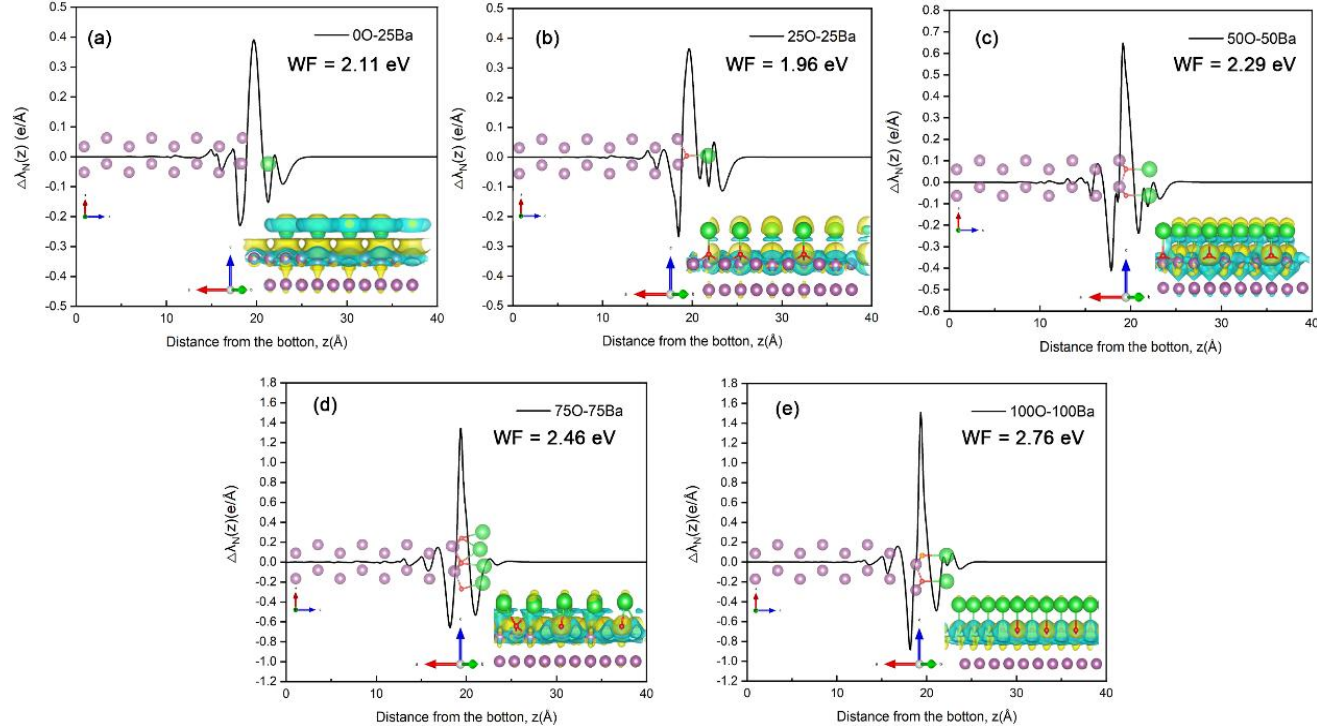


Fig. Work function of the addition of Sr in (a) BaO and (b) Ba_3WO_6



- The addition of 12.5%Sr could reduce the work function of BaO
- 8.3%Sr, The work function of Ba in Ba_3WO_6 is the lowest 2.415 eV



The curve of charge density difference $\Delta N(z)$ for the adsorption of Ba_xO_y on the Sc (002) surface with coverage of (a) 0%O-25%Ba, (b) 25%O-25%Ba, (c) 50%O-50%Ba, (d) 75%O-75%Ba and (e) 100%O-100%Ba

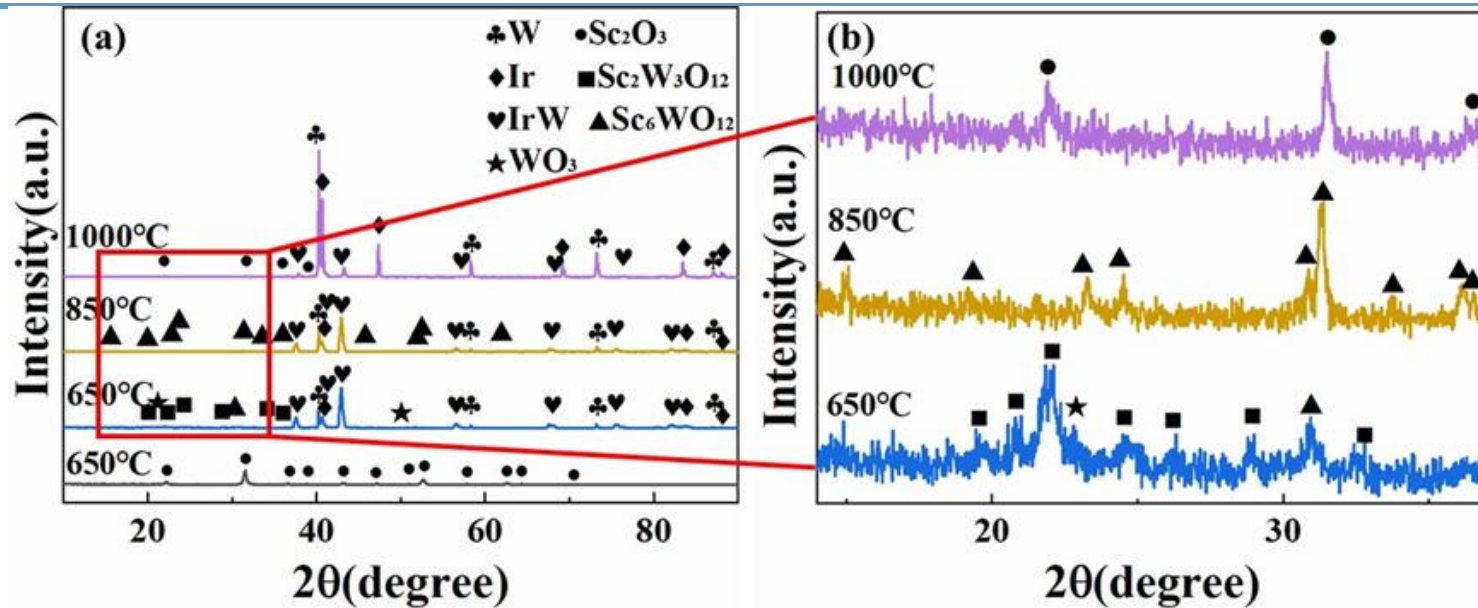
Table 1 The dipole moment $\Delta\mu_N$ of Ba_xO_y absorbed on Sc (002)

Sc (002)- Ba_xO_y	Work Function ϕ (eV)	Dipole moment changes $\Delta\mu_N$ (D)
0%O -25%Ba	2.11	0.309
25%O -25%Ba	1.96	0.351
50%O -50%Ba	2.29	0.289
75%O -75%Ba	2.46	0.215
100%O - 100%Ba	2.76	0.196

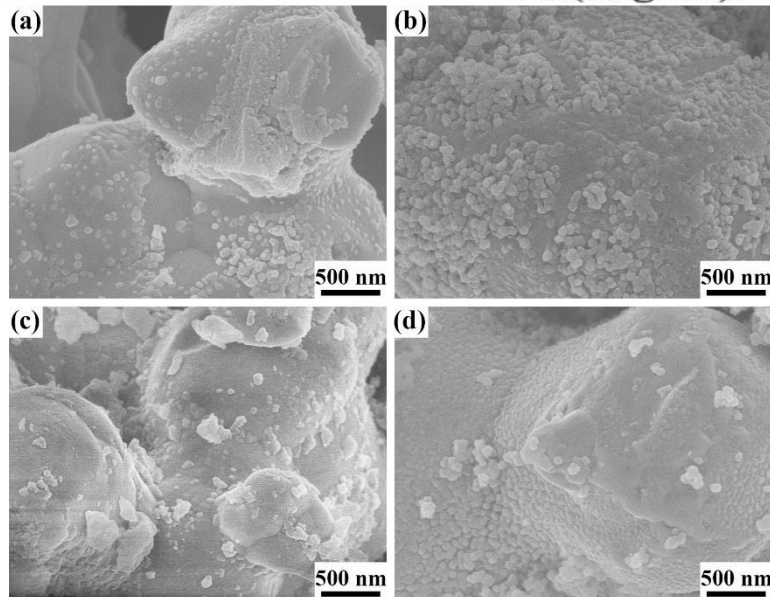


Ir-W matrix scandate cathode

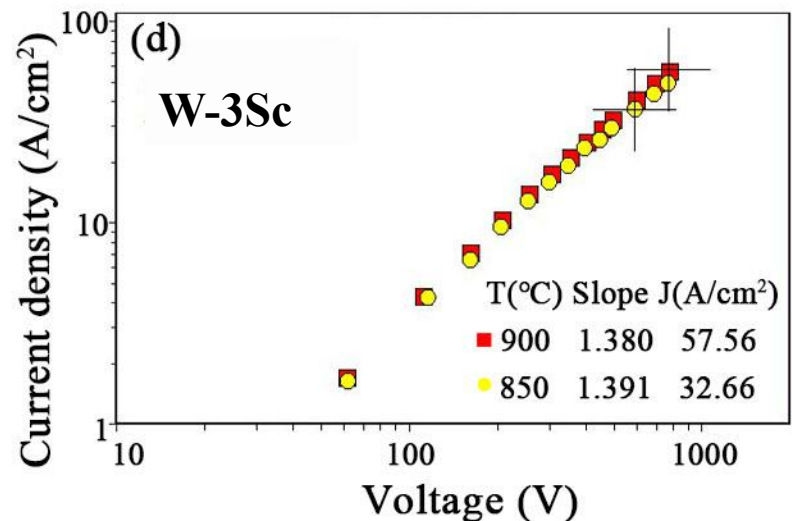
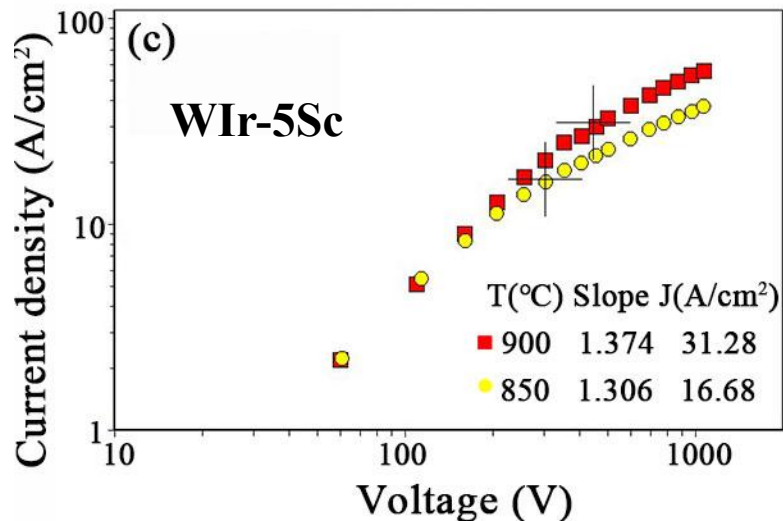
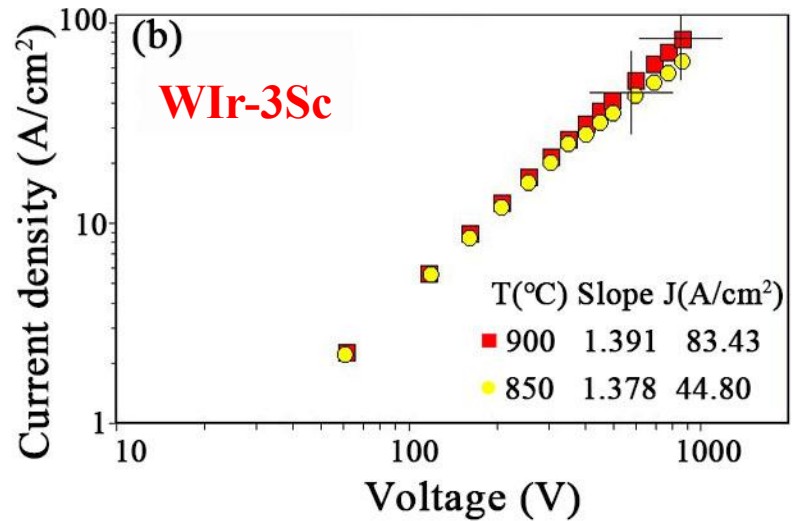
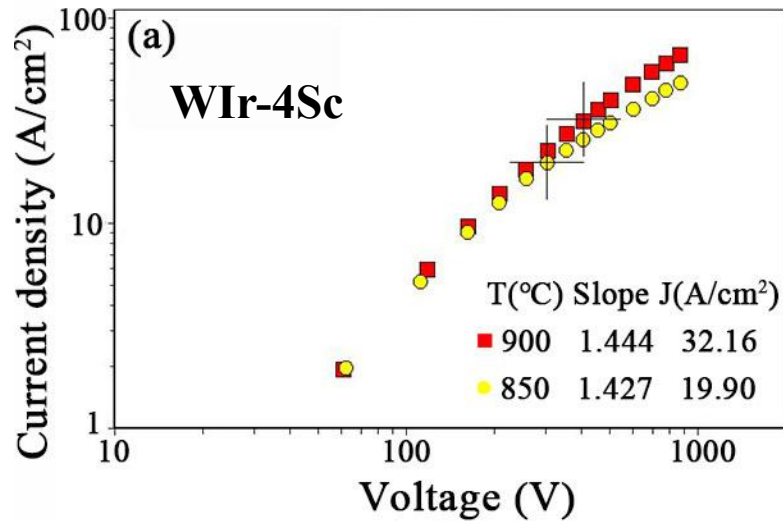
W-Ir scandate cathode



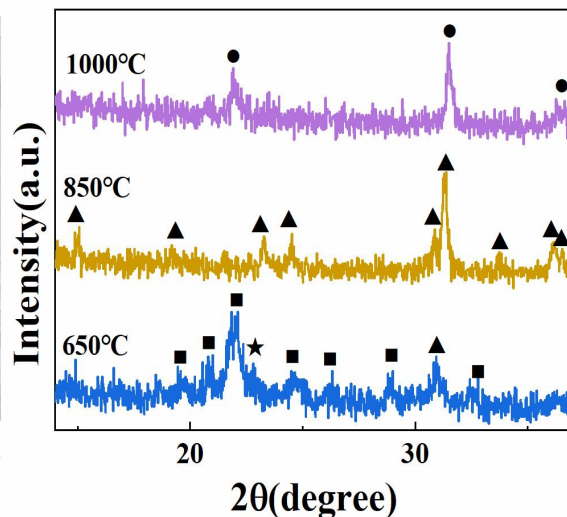
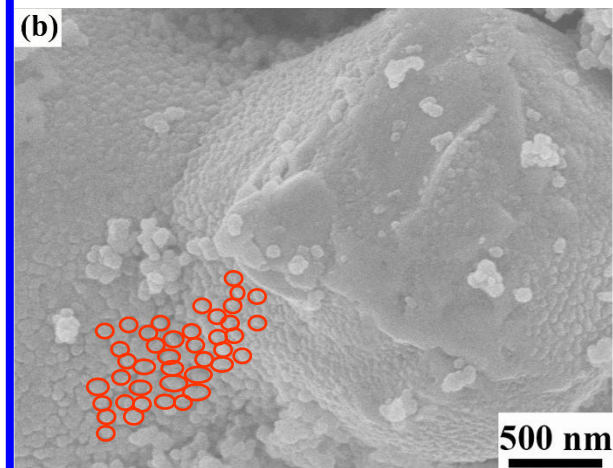
Sc_2O_3 nanoparticles grown on Ir and W powder surfaces synthesized by one step reduction



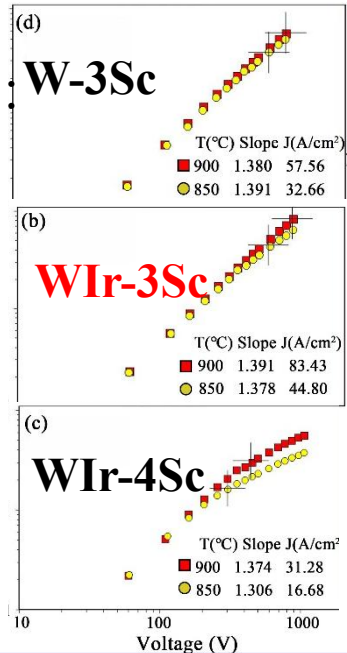
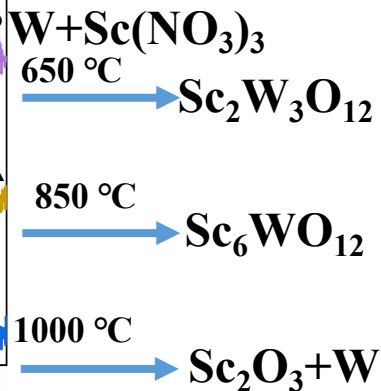
W-Ir scandate cathode



The electron emission current density could reach 83.43A/cm²

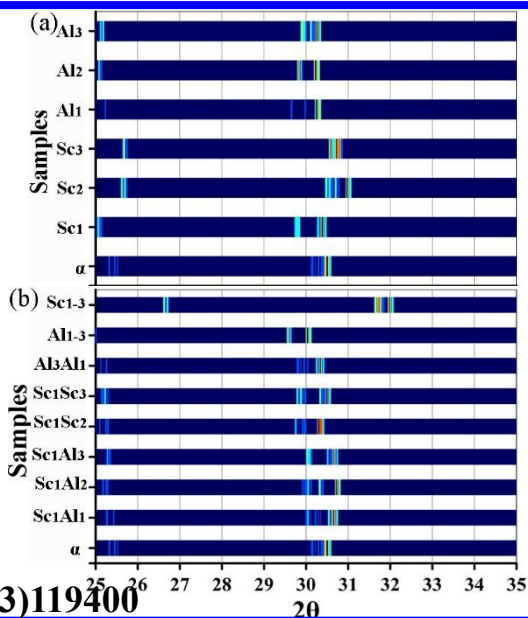
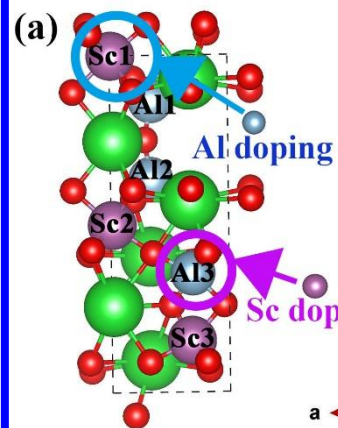


◆ Phase transformation

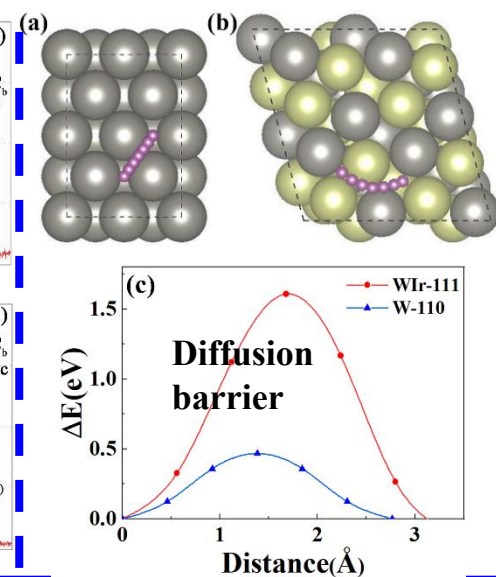
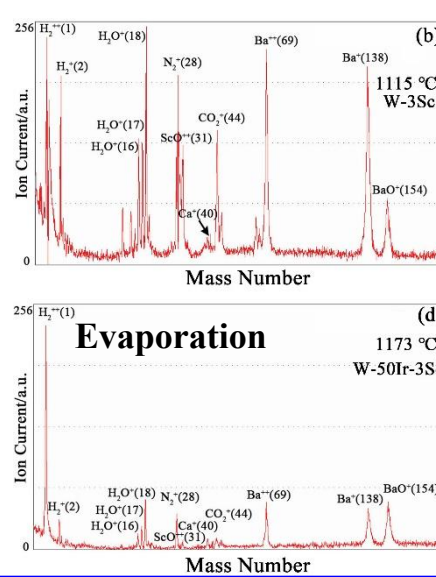


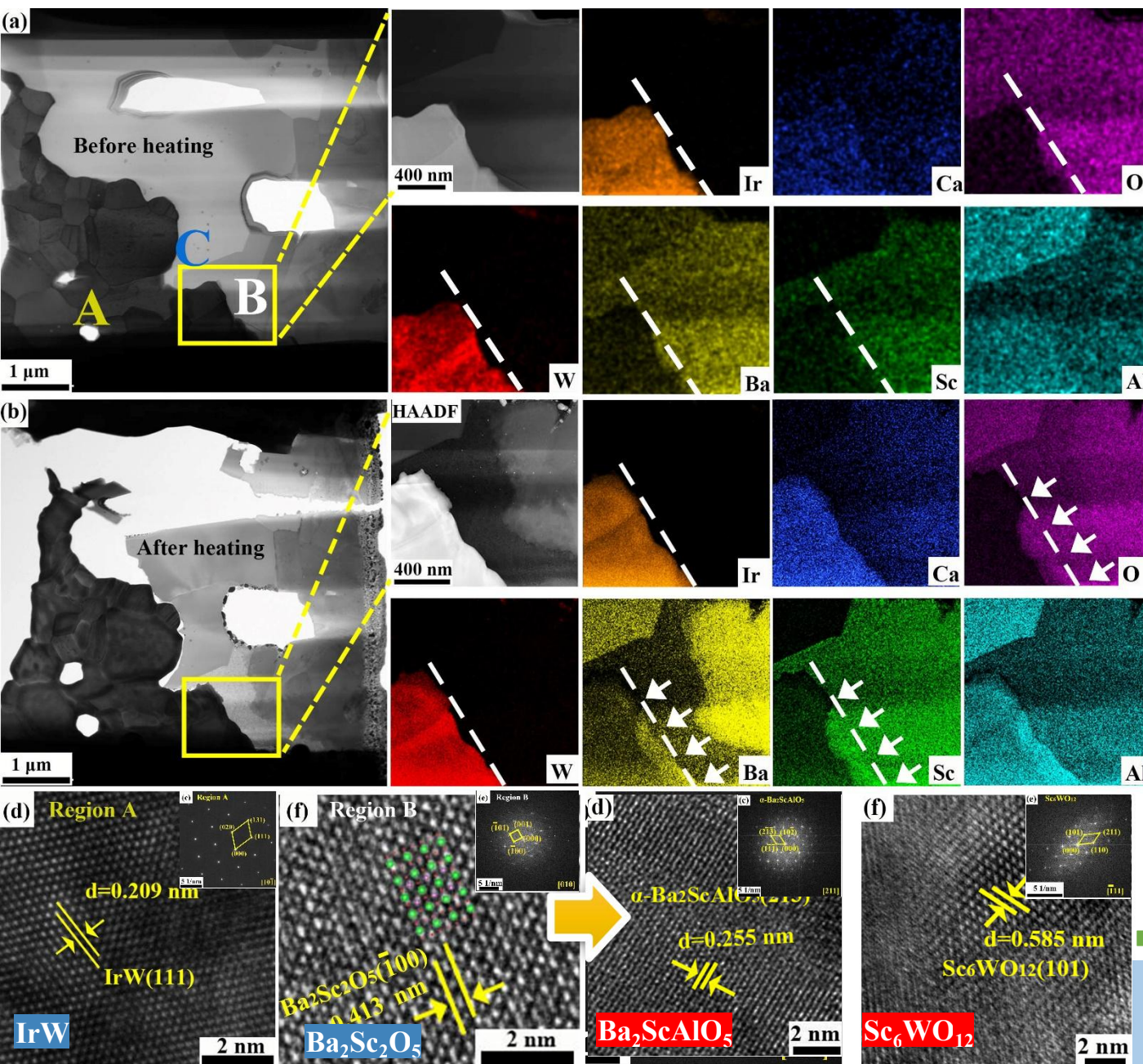
Powders grown on Ir and W surfaces of Sc₂O₃ nanoparticles synthesized in-situ in one step

◆ Sc: active phase

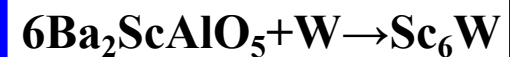
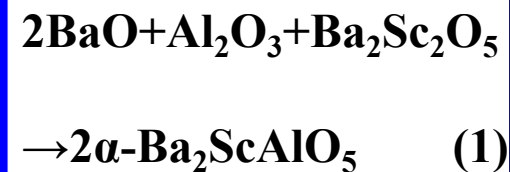


◆ Ir stabilizes active substance





◆ **Deduction :**



◆ **Design**

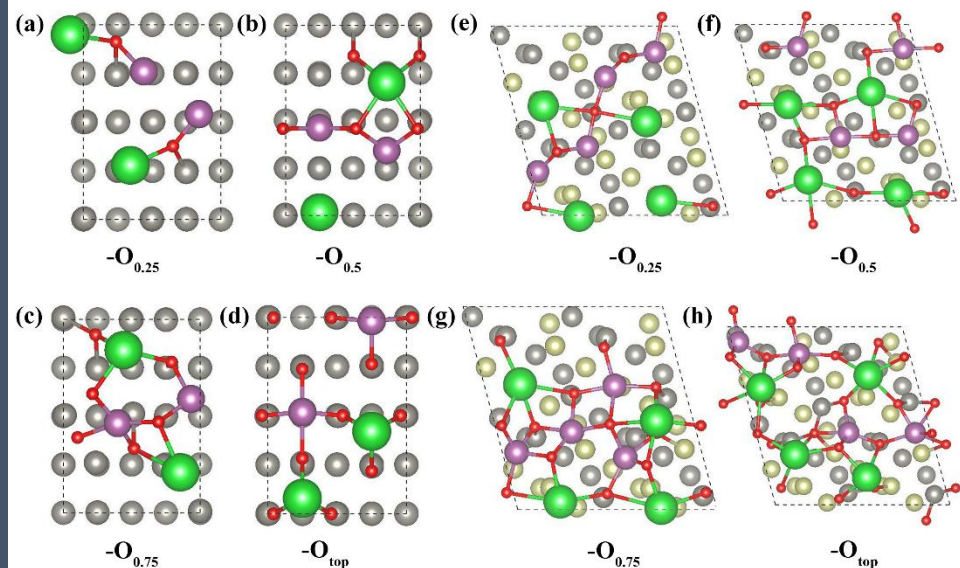
Emission mechanism

Characterization

Ir, Sc roles

Performance

Synthesized powder



$\text{Ba}_{0.25}\text{-Sc}_{0.25}\text{O}_x$ adsorbed on W and IrW

Table 3 Adsorption energy and formation energy

Crystal index	W (1 1 0)		IrW (1 1 1)	
	E_{ads}	ΔE_{F}	E_{ads}	ΔE_{F}
	/eV	/eV	/eV	/eV
$-\text{Ba}_{0.25}\text{-Sc}_{0.25}\text{-O}_{0.25}$	-4.89	-2.00	-5.88	-12.22
$-\text{Ba}_{0.25}\text{-Sc}_{0.25}\text{-O}_{0.5}$	-6.30	-8.73	-5.63	-19.16
$-\text{Ba}_{0.25}\text{-Sc}_{0.25}\text{-O}_{0.75}$	-6.97	-13.21	-8.58	-28.74
$-\text{Ba}_{0.25}\text{-Sc}_{0.25}\text{-O}_{\text{top}}$	-7.63	-12.26	-8.41	-29.47

◆ **Adsorption energy** : $\text{IrW-Ba}_{0.25}\text{-Sc}_{0.25}\text{-O}_{0.75} < \text{W-Ba}_{0.25}\text{-Sc}_{0.25}\text{-O}_{\text{top}}$

◆ Transferred electrons of Sc(1.88 e) and Ba(1.45 e) in $\text{IrW-Ba}_{0.25}\text{-Sc}_{0.25}\text{-O}_{0.75}$ is higher than Sc(1.44 e) and Ba(1.2 e) in $\text{W-Ba}_{0.25}\text{-Sc}_{0.25}\text{-O}_{\text{top}}$, and **Sc plays a major role**

◆ **Ir** makes the interaction between the active substance and the matrix stronger, **stabilizes the active substance**, and improves the emission performance of the cathode.

1. $(\text{Ba}, \text{Sr}, \text{Ca})_4\text{Al}_2\text{O}_7$ could improve the emission property of the impregnated cathode with the addition of appropriate amount of Sr in the Ba,Ca aluminates.
2. The calculation shows that properly increasing the concentration of oxygen on the Sc surface can reduce the work function.
3. Ir-W scandate cathode could provide the emission current density of $83\text{A}/\text{cm}^2$ at 900°C .
4. More transferred electrons of Sc($1.88 e$) and Ba($1.45 e$) in the cathode could benefit for cathode performance. Ir stabilizes the active substance on the cathode surface.

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Thanks for your kind attention